

# Ab Initio Molecular Dynamics Simulations in Liquids and Glasses

## Basics and Applications

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# Merci à ...

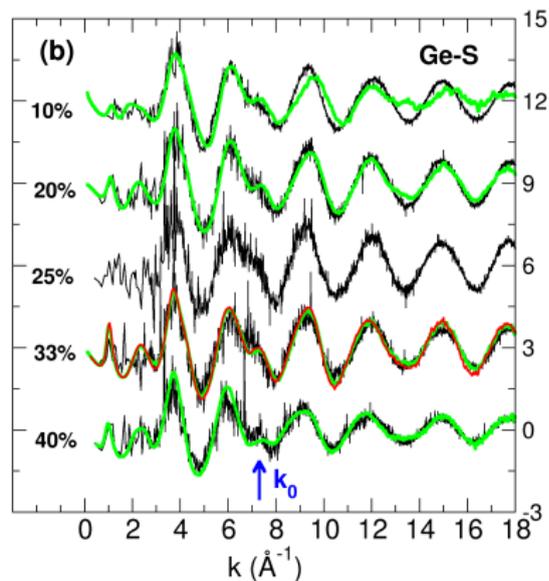
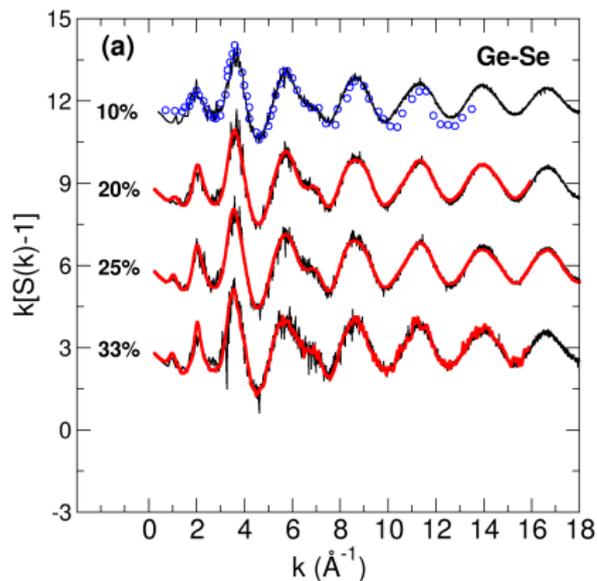


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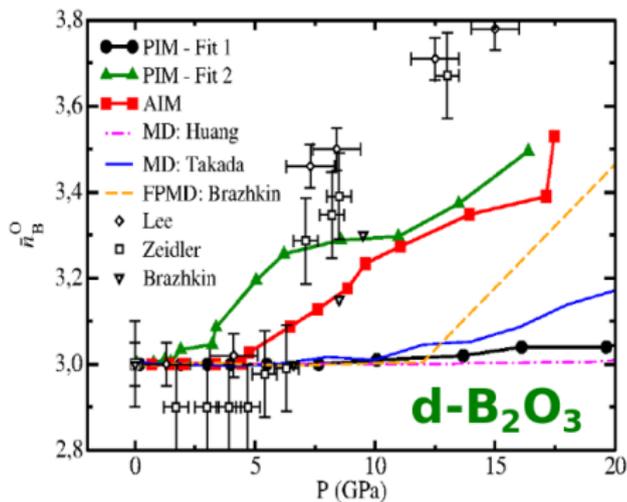
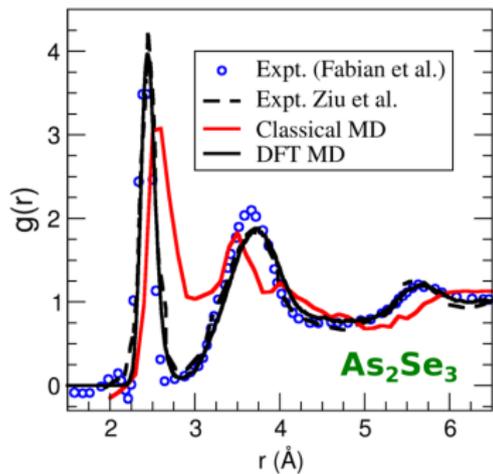
# Some personal thoughts and experiences



M. Micoulaut et al., PRB 88 (2013)

D. Chakraborty, P. Boolchand, M. Micoulaut, PRB 96 (2017)

# Some personal thoughts and experiences



A. Baroni et al., JCP 151 (2019)

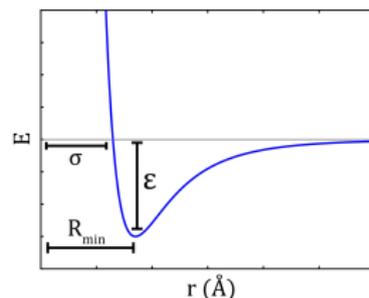
M. Bauchy, M. Micoulaut, JNCS 377 (2013)

# Some preliminary remarks...

- Classical force-fields carry their own and intrinsic limitations.

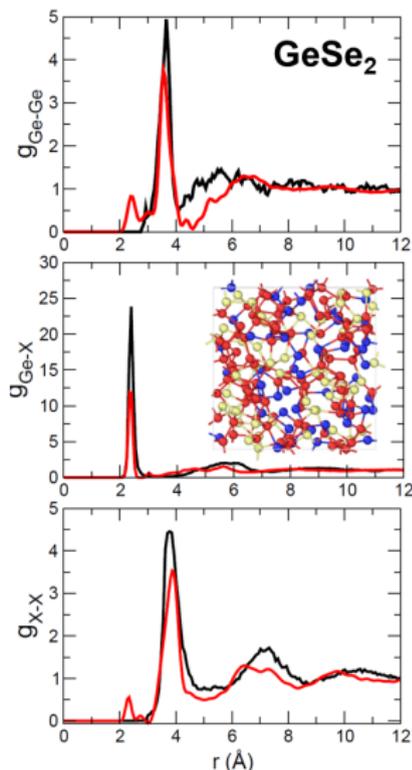
$$V_{ij}(r) = \frac{q_i q_j}{r} + A_{ij} e^{-r/\rho_{ij}} - \frac{C_{ij}}{r}$$

- Force-fields are not always available for a given system (composition) or thermodynamic condition (high pressure).
- No information available on the nature of the chemical bonding
- How to treat explicitly charge transfer in covalent systems (chalcogenides) ?
- Description of spectroscopic/electronic properties in relationship with atomic structure (Raman, IR, XPS,...)



# Some preliminary remarks...

- Some glass systems display broken chemical order.
- Classical force-fields do not account for such defects.
  - Ge-Te (phase change), Ge-Se or As-Se
  - HP-SiO<sub>2</sub>
- Systems with small band gaps undergo semiconducting to metallic transitions : As<sub>2</sub>Te<sub>3</sub>, GeTe,...



# Outline - Basics

- 1 Basics of ab initio simulations
  - Finding approximations
  - Solutions of the electronic Hamiltonian
- 2 Density functional theory
  - Exchange correlation
- 3 Implementation
  - Plane waves
  - Pseudopotentials
- 4 Ab initio molecular dynamics
  - Car-Parrinello MD
  - Fictive dynamics and recipes
- 5 Travaux pratiques
- 6 Conclusion

## FINDING APPROXIMATIONS

**The starting point** : the Schrödinger equation

$$H\Psi = E\Psi \quad (1)$$

for a system with  $N$  atoms and  $n$  electrons.

For large systems, there are basically two options :

- **Adiabatic approximation** :  $e^-$  move faster than the nuclei which can be considered as fixed.
- **Mean-field approximation** for the  $e^- - e^-$  interaction (density functional theory, DFT).

## FINDING APPROXIMATIONS

We write the Hamiltonian for molecules or solids. It contains the interaction and the kinetic energy of  $N$  atoms (nuclei of mass  $M$ ) at positions  $\mathbf{R}_I$  and  $n$  electrons of mass  $m$  at positions  $\mathbf{r}_i$ .

$$H = \sum_{i=1}^n \frac{\mathbf{P}_i^2}{2m} + \frac{1}{2} \sum_{i \neq j}^n \frac{e^2}{|\mathbf{r}_i - \mathbf{r}_j|} - \sum_{i=1}^n \sum_{I=1}^N \frac{Z_I e^2}{|\mathbf{r}_i - \mathbf{R}_I|} + \sum_{I=1}^N \frac{\mathbf{P}_I^2}{2M_I} + \frac{1}{2} \sum_{k \neq l}^N \frac{Z_k Z_l e^2}{|\mathbf{R}_k - \mathbf{R}_l|}$$

A useful form with standard identification

$$H = T_e(\{p\}) + V_{ee}(\{e\}) + V_{eN}(\{r\}, \{R\}) + T_N(\{P\}) + V_{NN}(\{R\})$$

Electronic kinetic E

e - e potential

e - nucleus potential

nuclear kinetic E

nucleus-nucleus potential

## FINDING APPROXIMATIONS

We solve the the Schrödinger equation :  $H\Psi_{(i,l)}(r, R) = E_{(i,l)}(r, R)$

**Adiabatic approximation** : Since  $M=2000 m_e$ , there are 2 different timescales involved : electrons and nuclei (ions)

- Vibrational excitations :  $\hbar\omega \simeq 10$  meV
- Electronic excitations :  $\hbar\omega \simeq 1$  eV

Since  $\tau_{ions} \gg \tau_e$ , on the timescale of electrons, ions can be considered as frozen.

**Consequence** : The wavefunction can be split in 2 parts

$$\Psi_{(i,l)}(r, R) = \chi_{(i,l)}(R)\Phi_e(r, R)$$

$$\left[ T_e + V_{ee} + V_{eN}(R) + V_{NN}(R) \right] \Phi_e(r, R) = E_e(R)\Phi_e(r, R)$$

$$\left[ T_n + E_e(R) \right] \chi_{(i,l)}(R) = E_{i,e}\chi_{(i,l)}(R)$$

## FINDING APPROXIMATIONS

- 1  $\Phi_e(R)$  is the ground state electronic wavefunction of the electronic Hamiltonian at fixed configuration (parameter)  $R$ .

$$\left[ T_e + V_{ee} + V_{eN}(R) + V_{NN}(R) \right] \Phi_e(r, R) = E_e(R) \Phi_e(r, R)$$

to obtain  $\Phi_e(R)$  and  $E_e(R)$  (**Born-Oppenheimer energy surface**).

- 2 Knowing  $E_e(R)$ , one solves

$$\left[ T_n + E_e(R) \right] \chi_{(i,l)}(R) = E_{i,e} \chi_{(i,l)}(R)$$

to obtain  $\chi_{(i,l)}(R)$  and  $E_{i,e}$ .

### Common approximations :

- At LT, harmonic approximation.  $\chi_{(i,l)}(R)$  only non-zero around :

$$E_e(R) \simeq E_e(R_0) + \frac{1}{2!} (R - R_0)^2 \frac{\partial^2 E_e(R)}{\partial R^2} + \dots$$

- At HT, nuclei behaves as a classical particle :  $M \frac{d^2 R}{dt^2} = - \frac{\partial E_e(R)}{\partial R}$

# Solutions of the electronic Hamiltonian

**Reducing complexity** : We want to compute  $\Phi_e(r, R) = \Phi_e(\mathbf{r}_1, \mathbf{r}_2, \mathbf{r}_3, \dots, \mathbf{r}_n, R)$

- Discretize space of each variable on a grid of  $N_G$  grid points. To define  $\Phi_e$ , one needs  $(N_G)^n$  values ! **Many body problem**
- Many properties can be reproduced by an independent electron approach (i.e. single particle). In this case, a single particle wavefunction needs only  $n \times N_G$  variables (Hartree-Fock).
- How do we construct an ab-initio independent electron approach of the **total energy** of the ground state ?
- Many static and quasistatic properties can be obtained as derivatives of the total energy :

$$\text{Force: } -\frac{\partial E(R)}{\partial \mathbf{R}}$$

$$\text{Dynamic matrix: } \frac{\partial^2 E(R)}{\partial \mathbf{R}_i \partial \mathbf{R}_j}$$

$$\text{Polarisation: } -\frac{\partial E(R)}{\partial \mathbf{E}_{\text{elec}}}$$

# Solutions of the electronic Hamiltonian

**Reminder from Hartree-Fock** : ab initio single particle approach for the total energy. Variational method with **exact** Hamiltonian and an **approximate** wavefunction.

$$E_{HF} = \min_{\Phi_{HF}} \langle \Phi_{HF} | H | \Phi_{HF} \rangle \geq E_{exact} \quad (2)$$

with

$$\Phi_{HF}(r_1, r_2, \dots, r_n) = \frac{1}{\sqrt{n!}} \begin{vmatrix} \psi_1(r_1) & \psi_2(r_1) & \dots & \psi_n(r_1) \\ \psi_1(r_2) & \psi_2(r_2) & \dots & \psi_n(r_2) \\ \dots & \dots & \dots & \dots \\ \psi_1(r_n) & \psi_2(r_n) & \dots & \psi_n(r_n) \end{vmatrix}$$

Single Slater determinant. Exact solution for an independent electron Hamiltonian with non-degenerate ground state.

## To do better, we change the Hamiltonian

- System with interacting electrons in a potential  $V_{ext}(r)$
- **Instead** : System with non-interacting electrons in a fictitious potential  $V'_{ext}(r) \neq V_{ext}(r)$  with same total energy and same electronic density.

## Inhomogeneous Electron Gas\*

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(Received 18 June 1964)

This paper deals with the ground state of an interacting electron gas in an external potential  $v(\mathbf{r})$ . It is proved that there exists a universal functional of the density,  $F[n(\mathbf{r})]$ , independent of  $v(\mathbf{r})$ , such that the expression  $E \equiv \int v(\mathbf{r})n(\mathbf{r})d\mathbf{r} + F[n(\mathbf{r})]$  has as its minimum value the correct ground-state energy associated with  $v(\mathbf{r})$ . The functional  $F[n(\mathbf{r})]$  is then discussed for two situations: (1)  $n(\mathbf{r}) = n_0 + \tilde{n}(\mathbf{r})$ ,  $\tilde{n}/n_0 \ll 1$ , and (2)  $n(\mathbf{r}) = \varphi(\mathbf{r}/r_0)$  with  $\varphi$  arbitrary and  $r_0 \rightarrow \infty$ . In both cases  $F$  can be expressed entirely in terms of the correlation energy and linear and higher order electronic polarizabilities of a uniform electron gas. This approach also sheds some light on generalized Thomas-Fermi methods and their limitations. Some new extensions of these methods are presented.

# Density functional theory

## STARTING FEATURES

- The electronic density is the basic variable.
- $N$  interacting electrons in an external potential  $V_{\text{ext}}(\mathbf{r})$  in a **unique** non degenerate GS:

$$n(\mathbf{r}) = N \int \int \int \Psi_{\text{GS}}^*(\mathbf{r}, \mathbf{r}_2, \dots, \mathbf{r}_N) \Psi_{\text{GS}}(\mathbf{r}, \mathbf{r}_2, \dots, \mathbf{r}_N) d^3\mathbf{r}_2 \cdot \mathbf{r}_3 \dots \mathbf{r}_N$$

- Hohenberg-Kohn theorem : It is shown that  $V'_{\text{ext}}(r) = V_{\text{ext}}(r) + \text{cst}$  where  $V'_{\text{ext}}(r)$  is the most general potential and  $n(\mathbf{r})$  is the GS density.

# Density functional theory

- **Definition :** The DFT functional is given by :

$$F[n] = \langle \Psi_{GS} | T_e^{[n]} + V_{ee}^{[n]} | \Psi_{GS} \rangle$$

with :  $N = \int n(\mathbf{r}) d^3\mathbf{r}$ . The energy of the system is given by :

$$E[n] = F[n] + \int n(\mathbf{r}) V_{ext}(\mathbf{r}) d^3\mathbf{r}$$

and minimized by the ground state density  $E[n_{GS}] = E_{GS}$ .

- **Kohn-Sham approach :** Given an interacting GS charge density  $n_{GS}(\mathbf{r})$  of a  $N$  electron system.
  - There exists just one external potential  $V_{KS}(\mathbf{r})$  for a non-interacting  $N$  electron system with GS state density  $n_{GS}(\mathbf{r})$
  - Defines a Kohn-Sham (single particle) Hamiltonian :

$$H_{KS} = \frac{P^2}{2m} + V_{KS}(\mathbf{r})$$

# Density functional theory

The solutions (eigenvectors  $|\Psi_{KS}^i\rangle$ , energies  $\epsilon_{KS}^i$ ) of the KS Hamiltonian satisfy :

$$n(\mathbf{r}) = 2 \sum_{j=1}^{N/2} \langle \Psi_{KS}^j | \mathbf{r} \rangle \langle \mathbf{r} | \Psi_{KS}^j \rangle$$

out of which 2 new quantities can be defined :

- The kinetic energy of the KS system

$$T_0[n] = 2 \sum_{j=1}^{N/2} \left\langle \Psi_{KS}^j \left| \frac{p^2}{2m} \right| \Psi_{KS}^j \right\rangle$$

It is not the kinetic energy part of the real interacting system but the fictitious non-interacting one !

- The Hartree Coulomb interaction ("classical" form)

$$E_H[n] = \frac{1}{2} \int \int d^3\mathbf{r} d^3\mathbf{r}' \frac{e^2 n(\mathbf{r}) n(\mathbf{r}')}{|\mathbf{r} - \mathbf{r}'|}$$

# Density functional theory

By definition, the **exchange-correlation** functional is :

$$E_{XC}[n] = F[n] - T_0[n] - E_H[n]$$

with the total energy given by :

$$\begin{aligned} E[n] &= T_0[n] + E_H[n] + E_{XC}[n] + \int n(\mathbf{r}) V_{\text{ext}}(\mathbf{r}) d^3\mathbf{r} \\ &= T_0[n] + E_{KS}[n] \end{aligned}$$

## Conclusion

Most of the difficult (and approximation) part is hidden in  $E_{XC}$  !  
The choice of  $E_{XC}$  dramatically impacts structural properties of the glass.

## EXCHANGE CORRELATION

We want to calculate :

$$E_{XC}[n] = \frac{1}{2} \int n(\mathbf{r}) d^3r \int d^3\mathbf{r}' \frac{n_{XC}(\mathbf{r}, \mathbf{r}' - \mathbf{r})}{|\mathbf{r} - \mathbf{r}'|}$$

with :

$$n_{XC}(\mathbf{r}, \mathbf{r}' - \mathbf{r}) = n(\mathbf{r}') \int d\lambda [g(\mathbf{r}, \mathbf{r}', \lambda) - 1]$$

involving the pair correlation function  $g(\mathbf{r}, \mathbf{r}', \lambda)$  of a system with density  $n(\mathbf{r})$  and electron-electron interaction  $\lambda$ .

- Classically, one would have  $P(\mathbf{r}, \mathbf{r}') \simeq n(\mathbf{r})n(\mathbf{r}')$  (Hartree-Coulomb energy). But this neglects the possibility of (spin) symmetry (Pauli **exchange interaction**).
- $n_{XC}$  is the **exchange-correlation hole**, a QM zone surrounding every electron in an interacting system that reduces the probability  $P(\mathbf{r}, \mathbf{r}')$  of finding other electrons within the immediate vicinity.

## EXCHANGE CORRELATION

### Approximations :

- Local density approximation (LDA) :

$$E_{XC}[n] \simeq E_{XC}^{LDA}[n] = \int n(\mathbf{r})\epsilon(n(\mathbf{r}))d^3\mathbf{r}$$

- $\epsilon(n(\mathbf{r}))$  is just a function of the local electronic density. Exchange-correlation energy/electron of a homogeneous interacting electron gas. Exact.
- Allows reproducing shell structure of atoms, bond lengths, vibrational frequencies.
- Generalized gradient approximation (GGA) :

$$E_{XC}^{GGA}[n] = \int d^3\mathbf{r} f\left(n(\mathbf{r}), \nabla n(\mathbf{r})\right)$$

## EXCHANGE CORRELATION

- $E_{XC}$  builds on reasonable electronic models, often broken up into exchange and correlation functionals :

$$E_{XC}[n] \simeq E_X[n] + E_C[n]$$

Several expressions for the correlation energy (fitted to results of accurate QMC calculations, e.g. uniform electron gas).

$$E_X^{GGA}[n] = \frac{4}{3} \left( \frac{3n(\mathbf{r})}{\pi} \right)^{1/3} - \sum_{\sigma} F(s_{\sigma}) n_{\sigma}^{4/3}(\mathbf{r}) d^3\mathbf{r}$$

with, e.g. :

$$F_{Becke} = \frac{\beta s_{\sigma}}{1 + 6\beta s_{\sigma} \sinh^{-1} s_{\sigma}}$$

- Various contributions. Complicated expressions. Usually referred by their abbreviations : Becke-Perdew-Wang (BPW91), BLYP, PBE, etc. ...
- Hybrid functionals (HF treatment of exchange energy)

# Density functional theory

## EXCHANGE CORRELATION

**Remember** : GGA provides better geometries for weak bonds which are too short in LDA.

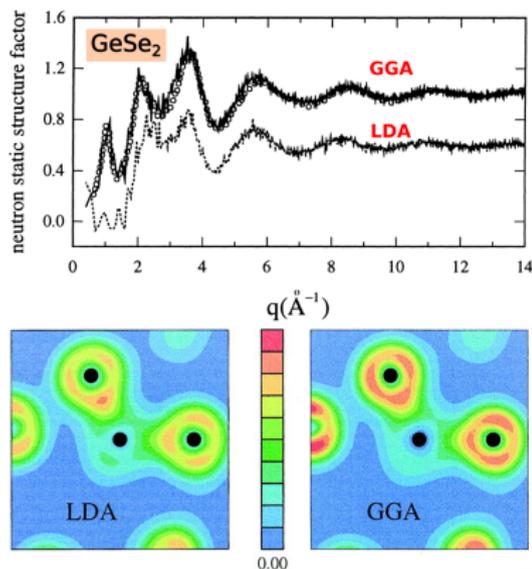
Solid	$a^{\text{LDA}}$ (Å)	$a^{\text{GGA}}$ (Å)	$a^{\text{expt}}$ (Å)
Na	4.05	4.20	4.23
NaCl	5.47	5.70	5.64
Al	3.98	4.05	4.05
Si	5.40	5.47	5.43
Ge	5.63	5.78	5.66
GeAs	5.61	5.76	5.65
Cu	3.52	3.63	3.60
W	3.14	3.18	3.16

Perdew et al. PRL 82 (1999)

# Density functional theory

## EXCHANGE CORRELATION

- Modelling of such XC functionals beyond the scope of this lecture.
- Many functionals available which are more or less appropriate for a particular glass/liquid system.
- Importance of GGA for the reproduction of intermediate range ordering in glass-forming liquids.



Massobrio, Pasquarello and Car, JACS 121 (1999)

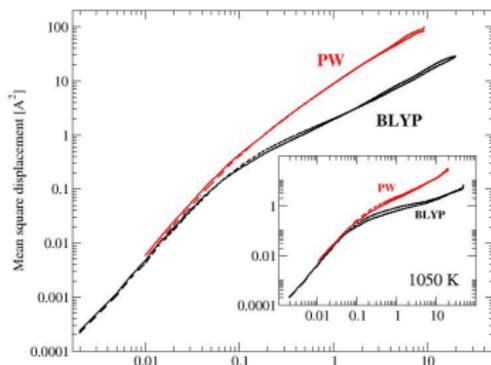
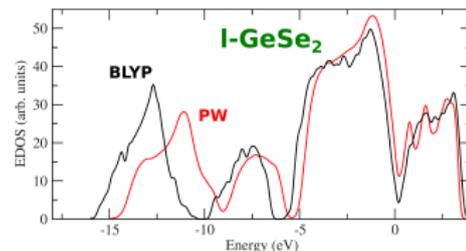
# Density functional theory

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Micoulaut, Vuillemier and Massobrio. PRB 79 (2009)

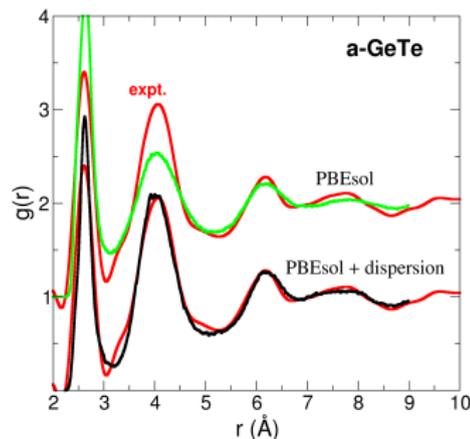
Micoulaut and Massobrio, JOAM 11 (2009) (2009)



# Density functional theory

## EXCHANGE CORRELATION

- Modelling of such XC functionals beyond the scope of this lecture.
- Many functionals available which are more or less appropriate for a particular glass/liquid system.
- Importance of GGA for the reproduction of intermediate range ordering in glass-forming liquids.
- Additional correction (dispersion forces)



Micoulaut et al. PRB 96 (2017)

$$E_{disp} = -s_6 \sum_i \sum_j \frac{C_{ij}}{R_{ij}^6} f_{damp}(R_{ij})$$

## EXCHANGE CORRELATION... IN SUMMARY

- Libraries of functionals routinely available.
- Aspects of accuracy problems encountered with classical force fields are also present in DFT calculations.
- Electronic treatment of the chemical bonding should not be overlooked.
- Check/compare your “candidate” functional with respect to experiments and results from other published functionals.

# Implementation

## PLANE WAVES

We are now trying to obtain the ground state (GS) energy and the electronic density

$$E_{GGA}^{GS} = \min_{\Psi_i^{KS}} H_{KS}(\Psi_i^{KS}, n(\mathbf{r}))$$

- In quantum chemistry, one often expands  $\Psi_i^{KS}$  on a finite basis (Hilbert space, tight-binding), e.g. with atomic orbitals:

$$|\Psi_i^{KS}\rangle = \sum_l \sum_n c_{l,n}^i |\Phi_{l,n}\rangle$$

- **Advantages:** smart basis and small size. Properties are calculated in terms of atomic orbitals.
- **Disadvantages :** Basis depends on atomic positions. No systematic way to increase the basis size (precision). Delocalized states in an empty region are hard to describe (e.g. crystal vacancy).

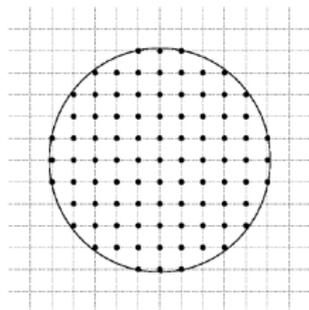
## PLANE WAVES

**Plane wave basis set** : Alternative for periodic systems : the basis functions of  $\Psi_i^{KS}$  can be expanded as plane waves.

$$\Psi_i^{KS} \propto \sum_{\mathbf{G}} c_i(\mathbf{G}) e^{i\mathbf{G}\cdot\mathbf{r}}$$

with  $\mathbf{G}$  a vector of the reciprocal lattice.

$$\mathbf{G} = \mathbf{i} \cdot \frac{2\pi}{L_x} + \mathbf{j} \cdot \frac{2\pi}{L_y} + \mathbf{k} \cdot \frac{2\pi}{L_z}$$



- Infinite sum is the FT of the wavefunction.
- **Advantages:** Basis covers all space uniformly. Does not depend on positions (weak bonds), fast evaluation (FFT). Improvement related to a cutoff value  $E_{cutoff}$ .
- **Disadvantages:** Large number of basis needed. Wavefunction in the chemical bond region varies too rapidly (pseudopotentials).

# Implementation

## PLANE WAVES A crucial parameter : $E_{cutoff}$

- Kinetic energy associated with plane waves (PW,  $T_0 \propto 1/2\partial_x^2$ ):

$$E_{cutoff} = \frac{1}{2} G_{max}^2$$

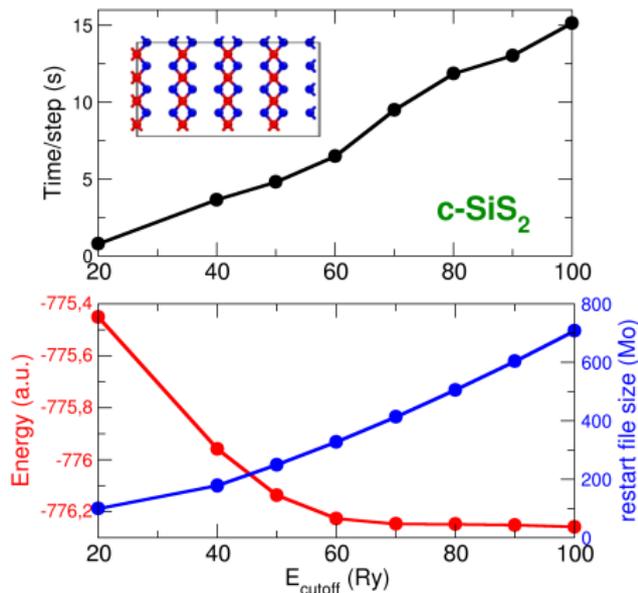
- Volume of the sphere containing all PW, volume occupied by a single PW :

$$V = \frac{4\pi}{3} G_{max}^3 \quad V_{PW} = \frac{2\pi}{\Omega}$$

- Number of PWs :

$$N_G \propto \Omega E_{cutoff}^{3/2}$$

Properties converge with increasing cutoff value !



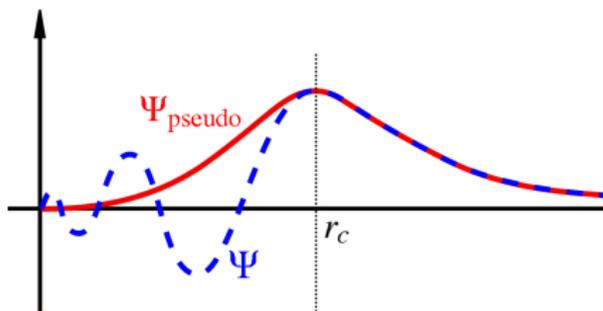
## PSEUDOPOTENTIALS

- Concept related to replacing the effects of the core electrons with an effective (pseudo) potential. **Just the outer shell electrons participate in the chemical bonding.**
- The pseudopotential generation procedure starts with the solution of the atomic problem (all electrons) using the Kohn-Sham approach.
- Once the KS orbitals are obtained, one makes an arbitrary distinction between valence and core states.
- The core states are assumed to change very little due to changes in the environment so their effect is replaced by a model potential derived in the atomic configuration and it is assumed to be transferable.
- Outside the core region, the pseudo-wavefunction coincides with the all electron calculation.

# Implementation

## PSEUDOPOTENTIALS

- General conditions for pseudo-potential generation
  - 1 All-electron and pseudo eigenvalues agree for the reference configuration.
  - 2 AE and PS wavefunctions agree beyond a certain cutoff value,  $r_c$ .
- Two opposing considerations :
  - 1 Good transferability small  $r_c$ .
  - 2 Large  $r_c$  smoother pseudopotentials.



A good pseudopotential is one that strikes a balance between these two constraints.

## AB INITIO MOLECULAR DYNAMICS

We would now like to follow the positions  $\mathbf{R}$  of the ions with time, in order to obtain the dynamics of liquids and glasses or to obtain vibrational dynamics (VDOS, IR,...). Two possible approaches :

- **Minimization technique:** Integration of the ion motion with  $\Delta t$  fixed by the largest ion frequency. After each ion move, re-minimization of  $\Psi$ . **Born-Oppenheimer MD**

$$m_i \ddot{\mathbf{R}}_i = - \frac{\partial}{\partial \mathbf{R}_i} \min_{\Psi} \langle \Psi | H | \Psi \rangle$$

- **Fictitious ion dynamics:** Separation of the dynamics into ion (classical) and electronic (fictitious) dynamics. Newton's equation of motion for the ions is solved. **Car-Parrinello MD**

## Unified Approach for Molecular Dynamics and Density-Functional Theory

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(Received 5 August 1985)

We present a unified scheme that, by combining molecular dynamics and density-functional theory, profoundly extends the range of both concepts. Our approach extends molecular dynamics beyond the usual pair-potential approximation, thereby making possible the simulation of both covalently bonded and metallic systems. In addition it permits the application of density-functional theory to much larger systems than previously feasible. The new technique is demonstrated by the calculation of some static and dynamic properties of crystalline silicon within a self-consistent pseudopotential framework.

PACS numbers: 71.10.+x, 65.50.+m, 71.45.Gm

## CAR PARRINELLO MOLECULAR DYNAMICS

**Reminder** : Equations of motions from classical MD for  $N$  atoms with position  $\mathbf{R}_i$

$$L(\mathbf{R}_i, \dot{\mathbf{R}}_i) = T_0(\dot{\mathbf{R}}_i) - V(\mathbf{R}_i)$$

From the momentum  $\mathbf{p}_i = \partial L / \partial \dot{\mathbf{R}}_i$  and force  $\mathbf{F}_i = \partial L / \partial \mathbf{R}_i$ , one recovers Newton's equations from the Euler-Lagrange equation as :

$$\frac{d}{dt} \frac{\partial L}{\partial \dot{\mathbf{R}}_i} = \frac{\partial L}{\partial \mathbf{R}_i} \quad \Rightarrow \quad \mathbf{p}_i = -\frac{\partial V}{\partial \mathbf{R}_i} \quad \Rightarrow \quad m_i \ddot{\mathbf{R}}_i = \mathbf{F}_i$$

## CAR PARRINELLO MOLECULAR DYNAMICS

Postulated Lagrangian by Car and Parrinello (1985):

$$L = \sum_I \frac{1}{2} M_I v_I^2 + \sum_i \frac{1}{2} \mu_i \langle \dot{\Psi}_i | \dot{\Psi}_i \rangle - E[\{\Psi\}, \{\mathbf{R}\}] + \sum_{ij} \Lambda_{ij} (\langle \Psi_i | \Psi_j \rangle - \delta_{ij})$$

nuclear kinetic energy      electron "kinetic energy"      potential energy      orbital orthogonality (L.M)

From Lagrangian mechanics, one obtains the equations of motion :

$$\frac{d}{dt} \frac{\partial L}{\partial \mathbf{v}} = \frac{\partial L}{\partial \mathbf{R}} \quad \Rightarrow \quad M_I \ddot{\mathbf{R}} = - \frac{\partial V}{\partial \mathbf{R}}$$

$$\frac{d}{dt} \frac{\partial L}{\partial \langle \dot{\Psi}_i |} = \frac{\partial L}{\partial \langle \Psi_i |} \quad \Rightarrow \quad \mu_i |\ddot{\Psi}_i\rangle = - \frac{\partial V}{\partial \langle \Psi_i |} + \sum_j \Lambda_{ij} |\Psi_j\rangle$$

Decoupling between a "real" ion dynamics (mass  $M_I$ ) and the fictitious electronic dynamics (mass  $\mu_i$ ).

## THE CAR PARRINELLO MOLECULAR DYNAMICS RECIPE

- 1 For a given set of  $\{\mathbf{R}_I\}$ , within DFT minimize  $V[\Psi_{KS}]$  to obtain  $\Psi_{KS}^{GS}$ .

$$V(\{\mathbf{R}_I\}) = \min_{\Psi} V[\{\Psi_{KS}\}, \{\mathbf{R}_I\}]$$

- 2 Compute the forces on the ions from :

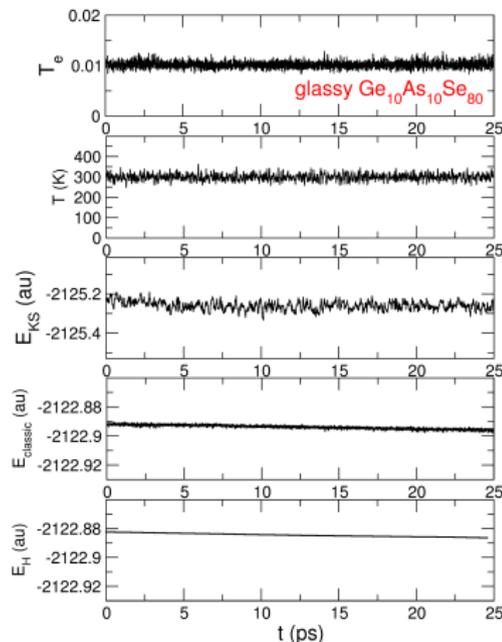
$$\mathbf{F}_I = -\frac{\partial}{\partial \mathbf{R}_I} V(\{\mathbf{R}_I\})$$

- 3 Move the ions with Newton's equation
- 4 Integration with Verlet algorithm to get a trajectory

$$\mathbf{R}_I(t + \Delta t) + \mathbf{R}_I(t - \Delta t) = 2\mathbf{R}_I(t) + \frac{\mathbf{F}_I}{M_I}(\Delta t)^2$$

## THE CAR PARRINELLO MOLECULAR DYNAMICS RECIPE

- CPMD simulations of 249 atom  $\text{Ge}_{10}\text{As}_{10}\text{Se}_{80}$
- $\Delta t = 5$  a.u. (0.24 fs),  $L = 17.98$  Å.
- $E_{\text{cutoff}} = 20$  Ry, BLYP,  $\mu = 2000$  a.u.
- Kinetic energy  $T_e$  of the electrons (electronic sub-system) remains small.
- Kohn-Sham energy  $E_{KS}$  equivalent to  $V(r)$  in classical MD.
- $E_{\text{classic}} = E_{KS} + \text{ionic kinetic energy}$
- $E_H = E_{\text{classic}} + T_e$ , the conserved parameter in CPMD
- Total energy  $E_H$  conserved.



# Ab initio molecular dynamics

## CAR PARRINELLO VERSUS BORN-OPPENHEIMER

Simulations performed on liquid (1000 K)  $93 \text{ B}_2\text{O}_3 - 7 \text{ Li}_2\text{O}$  ( $N=146$  atoms)

Method	Time step (a.u.)	Convergence (a.u.)	Conservation (a.u./ps)	Time (s)
CP	5	-	$6 \times 10^{-8}$	30 270
CP	7	-	$1 \times 10^{-7}$	21 648
CP	10	-	$3 \times 10^{-7}$	15 088
BO	10	$10^{-6}$	$1 \times 10^{-6}$	155 492
BO	50	$10^{-6}$	$1 \times 10^{-6}$	38 709
BO	100	$10^{-6}$	$6 \times 10^{-6}$	21 088
BO	100	$10^{-5}$	$1 \times 10^{-5}$	15 558
BO	100	$10^{-4}$	$1 \times 10^{-3}$	9 934

Table: Timing in CPU seconds and energy minimization for 1 ps trajectory

## FICTITIOUS MASS OF THE ELECTRON (ADIABATICITY PARAMETER)

- **Meaning:**  $\mu$  is non-physical parameter controlling the time-scale of the "classical" CP electronic dynamics.
- An optimal value permits to stay near the Born-Oppenheimer surface: an diabaticity separation between electronic and ionic motion.
- The electron "subsystem" must be kept "cold" (thermostats). In this case, the e-system remains close to GS and  $T_e$  is small.
- To avoid energy exchange between the nuclear and orbital degrees of freedom, **power spectra should not overlap.**

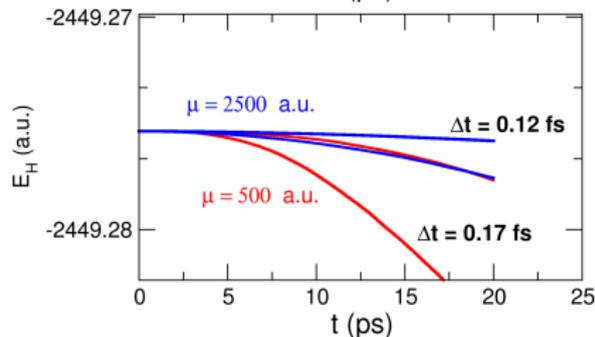
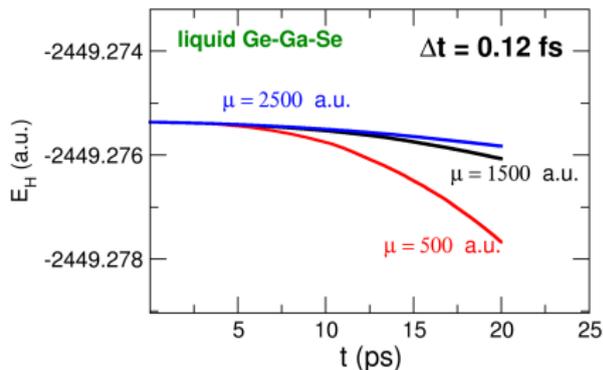
$$\omega_{ij} = \sqrt{\frac{2(E_i^{KS} - E_j^{KS})}{\mu}} \quad \Rightarrow \quad \omega_e^{min} \simeq \sqrt{\frac{2(E_{lumo} - E_{homo})}{\mu}} = \sqrt{\frac{2E_{gap}}{\mu}} \gg \omega_{ions}^{Max}$$

## FICTITIOUS MASS OF THE ELECTRON (ADIABATICITY PARAMETER)

- $E_{cutoff}$ ,  $E_{gap}$ ,  $\mu$  and timestep  $\Delta t_{max}$  are related.
- The highest frequency for the subsystem also depends on  $\mu$  and constrains the maximum allowed MD timestep:

$$\Delta t_{max} \propto \frac{1}{\omega_e^{Max}} \propto \sqrt{\frac{\mu}{E_{cutoff}}}$$

- Commonly used values :  
 $\mu=300-2000$  a.u.  $\Delta t=0.1-0.2$  fs.



## HOW TO GET STARTED

- 1 What you need to specify to run a DFT computation.
  - Coordinates
  - Exchange-correlation functionals: BLYP, PW, HS06, PBE...
  - Corresponding pseudopotentials
- 2 Packages
  - CPMD or Quantum Espresso (CP2K)
  - ABINIT
  - VASP
  - Wikipedia : List of quantum chemistry and solid-state physics software
- 3 Lectures
  - Detailed course on quantum chemistry, DFT, and ab initio methods.
  - CECAM tutorials [www.cecam.org](http://www.cecam.org)
- 4 Analysis tools (similar to classical MD).

## CONCLUSION

- Ab initio methods allow to investigate glass properties using atomic number as only input.
- Theoretical background provided by density functional theory (DFT).
- Additional approximations are needed to make DFT useful and applicable.
- Combination of DFT with dynamics through different techniques (minimisation, fictitious dynamics).
- **Next:** Application of ab initio methods to liquids and glasses